Large Seebeck coefficients in iron-oxypnictides: a new route towards n-type thermoelectric materials

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The iron-oxypnictide compounds, recently reported as a new class of superconductors when appropriately doped, exhibit large Seebeck coefficients, of the order of –100 µV/K, while keeping good electrical conductivity. Their power factor shows a peak at low temperatures, suggesting possible applications of these materials in thermoelectric cooling modules in the liquid nitrogen temperature range.

1 Introduction

Rare-earth (RE) iron arsenide oxides (REFeAsO) were initially reported by Quebe et al. [1]. Single crystal analysis showed that these compounds crystallize in a P4/nmm structure, containing two REFeAsO units per unit cell, which consists of alternating iron arsenide and rare-earth oxide layers. Little work was published on these materials until it was found recently that, when doped with fluorine on the oxygen site, electron-doped LaAsFeO becomes a superconductor (SC) with \( T_c = 26 \) K [2]. Following these surprising findings, it has been noted that the superconductivity is induced as well by hole doping [3]. By adjusting the lattice volume via rare-earth replacement, higher critical temperatures were reported [4]. While a huge effort is presently devoted on experimental and theoretical aspects, no clear understanding of both superconducting and normal state properties is achieved yet. In this report, we present evidence of a large Seebeck coefficient in these materials coupled with good electrical conductivity. This leads to promising thermoelectric power factors that could enable the use of these materials in thermoelectric modules in refrigeration applications around liquid nitrogen temperatures.

2 Experimental

REASFeO compounds were synthesized, as originally reported, in closed silica tubes heated at high temperatures, backfilled with pure Ar at about 200 mbar pressure (at ambient temperature). We will discuss in the following the synthesis of the compound SmAsFeO\(_{0.8}F_{0.2}\) (SmAsFeO(F)); other compounds with Nd or La were synthesized in a similar manner and some illustrative results are shown here. SmAs alloys were obtained from Sm metal and As chips following a procedure suggested by Hiscocks and Mullin [5]. The obtained SmAs alloy was mixed with Fe\(_3\)O\(_5\), Fe and with FeF\(_3\) in the stoichiometric ratios. The powdered compounds were thoroughly mixed and pressed as small bars. Heating of these bars wrapped in tantalum foil, sealed in silica tubes, at 1150–1160 °C for more than 40 hours, yielded well crystallized samples. X-ray diffraction patterns were obtained with a Panalytical X’Pert with a Ge(111) incident monochromator and a X’celerator detector. Data were refined by the Rietveld method, by using the GSAS package [6]. Electrical resistivity was measured using a DC four wires method in a closed cycle cryostat, from 300 K to about 30 K. The thermoelectric power was measured by a differential method by using the slope of the \( \Delta V \) versus \( \Delta T \) curve with gradients up to about 0.2 K/mm, by using a laboratory made system. All transport measurements were performed in a direction perpendicular to the pressing direction. Magnetic properties were measured in a commercial Quantum Design MPMS, under various fields, down to 2 K.
3 Results and discussion

We report here large thermoelectric power factors, for some of the compounds studied in this family, that indicate potential applications for these materials in thermoelectric cooling modules.

X-ray powder diffraction on a SmAsFeO(F) sample obtained after one heat treatment evidences, besides the P4/nmm peaks, the presence of minor impurities, SmOF and SmAs, estimated at under 10%. The bulk magnetic susceptibility measured after Zero Field Cooling (ZFC) in $H = 5$ G shows a sharp superconducting transition with $T_c = 48$ K as plotted in the inset of Fig. 1. The superconducting fraction of about 90% demonstrates the quality of the SmAsFeO phase in this sample. DC resistivity (Fig. 1) and Seebeck coefficient (Fig. 2a) confirmed this behaviour, showing a clear SC transition at about 50 K with a resistivity of about $2 \times 10^{-3} \Omega \text{cm}$ before the transition. A second thermal treatment leads to a slight improvement of the critical temperature and a significant decrease of the resistivity, by one order of magnitude. This treatment induced a slight decrease of the cell volume together with a decrease of the SmOF impurity fraction. These findings are consistent with an increase of the fluorine concentration in the SmAsFeO(F).

The Seebeck coefficient measured after the first heat treatment reaches a maximum value of $-90 \mu \text{V K}^{-1}$ at about 80 K, with a well defined peak. The second thermal treatment decreases the absolute value to about $-60 \mu \text{V K}^{-1}$ probably following the increase of the electron concentration with doping of fluorine on the oxygen site.

We also obtained large Seebeck coefficients for Nd samples ($-100 \mu \text{V K}^{-1}$ at 90 K) and for La samples ($-130 \mu \text{V K}^{-1}$). A similar value was reported for a LaFeAsO(F) sample [7]. Up to now, the best thermoelectric performance in the 80–100 K temperature range is found in single crystalline Bi–Sb alloys, with a thermoelectric figure of merit $ZT$ reaching about 0.5 ($ZT = S^2 T/\rho \lambda$ with $S$ the Seebeck coefficient, $\lambda$ the thermal conductivity and $\rho$ the electrical resistivity). However, these single crystals are both hard to produce and quite brittle, which make them unsuitable for practical applications. Therefore, our results should rather be compared to those obtained in polycrystalline Bi–Sb alloys. As it can be seen in Fig. 2a, the order or magnitude of the Seebeck coefficient of iron-oxypnictides is very similar to the best values obtained in polycrystalline Bi–Sb alloys [8], and reaches about $-130 \mu \text{V K}^{-1}$ around 100 K. These large Seebeck coefficient values might be related to electron correlations or to a 2D electron confinement in FeAs layers.

Concerning the electrical resistivity, it is moderately higher in iron-oxypnictides than in Bi–Sb alloys, leading to a lower thermoelectric power factor $S^2/\rho$ (Fig. 2b): the power factor reaches 1 mW m$^{-1}$ K$^{-2}$ in SmFeAsO$_{0.8}$F$_{0.2}$, which is about 7 times lower than in Bi$_{0.88}$Sb$_{12}$.
However, it must be emphasized that the power factor measured in these compounds is not so far from that of Bi–Sb alloys, as our samples are poorly densified, and that much lower electrical resistivity values have been reported for single crystals. This should lead to much higher power factor values. For instance, a resistivity of $\rho \sim 4 \times 10^{-4} \Omega \text{cm}$ has been reported at 100 K in single crystals of Nd compounds [9], and assuming a similar Seebeck coefficient, this would correspond to a power factor of 2.5 mW m$^{-1}$ K$^{-2}$ which is about the same magnitude as in Bi$_{88}$Sb$_{12}$. On the other hand, in the same REFeAsO family, it has been reported that hole doping on the rare-earth oxide layer can yield p-type samples [3]. This would be of fundamental interest for applications, since the thermoelectric properties of the known p-type compounds are rather poor at low temperature. Therefore, the iron-oxypnictide materials appear to be very promising for thermoelectric cooling applications in the liquid nitrogen temperature range.

References